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Conductivity mapping of graphene on polymeric films by terahertz time-domain spectroscopy

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Abstract: Fast inline characterization of the electrical properties of graphene on polymeric substrates is an essential requirement for quality control in industrial graphene production. Here we show that it is possible to measure the sheet conductivity of graphene on polymer films by terahertz time-domain spectroscopy (THz-TDS) when all internally reflected echoes in the substrate are taken into consideration. The conductivity measured by THz-TDS is comparable to values obtained from four point probe measurements. THz-TDS maps of 25x30 cm² area graphene films were recorded and the DC conductivity and carrier scattering time were extracted from the measurements. Additionally, the THz-TDS conductivity maps highlight tears and holes in the graphene film, which are not easily visible by optical inspection.

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be able to measure the electrical properties non-destructively on other substrates than Si, such as on polymeric films would also be beneficial for process optimization of graphene transfers, as for instance polymeric films. The ability to map the key electrical properties of graphene on Si substrate to be highly resistive [7,8].

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Introduction

As chemical vapor deposition growth (CVD) of graphene is reaching industrial maturity, reliable large-area characterization of the electrical properties of graphene on a wide range of substrates is becoming a necessary requirement [1]. Terahertz time-domain spectroscopy (THz-TDS) has proven useful as a fast non-contact method for extracting the conductivity, carrier scattering time, and mobility of graphene on highly resistive silicon (Si) substrates through transmission-mode measurements [2–6]. Recently, the conductivity of graphene on Si has also been extracted from reflection-mode THz-TDS, which avoids the requirement of the Si substrate to be highly resistive [7,8].

For applications where the substrate needs to be mechanically flexible, it is necessary to be able to measure the electrical properties non-destructively on other substrates than Si, such as for instance polymeric films. The ability to map the key electrical properties of graphene on polymeric films would also be beneficial for process optimization of graphene transfers,
since most transfer methods utilize a polymer as handling layer during the transfer process [9–16].

Here, we use THz-TDS to measure the electrical properties of graphene on polyethylene terephthalate (PET) films, which are widely used as substrates for graphene devices. We show that the conductivity extracted from THz-TDS measurements is comparable to measurements conducted by four point probe (4pp) measurements. In order to show the amenability of THz-TDS as a characterization tool for large-scale sheet-based or roll-to-roll processing, we present conductivity maps of 25x30 cm² area graphene on PET. Additionally, the THz-TDS measurements highlight large defects such as tears and scratches in the graphene layer on polymeric substrates, which are difficult to assess by optical microscopy, in particular on transparent substrates.

2. Methods

Graphene was grown on copper foil by chemical vapor deposition (CVD) in a low pressure tube furnace system (40 cm diameter, 300 cm length) equipped with a distributed multifunction chamber system which can precisely control graphene growth parameters such as temperature, vacuum, and gas concentration. Prior to growth, the 25 μm thick copper foil was washed subsequently with acetone, ethanol and deionized (DI) water to remove any organic impurities. The cleaned copper foils were immersed in 1 wt% sulfuric acid for 3 minutes to remove surface oxides and then thoroughly rinsed with DI water. The copper foils were cut to 25x30 cm², placed in the quartz tube chamber and annealed at 1000 °C with 1000 sccm Ar and 500 sccm H₂ to remove residuals on the copper surface. During the CVD growth, the H₂ gas was kept at 500 sccm and CH₄ was added at 100 sccm for 30 minutes. After growth, CH₄ and H₂ were turned off and the copper/graphene was auto-delivered out under Ar gas protection.

The as-grown graphene on copper was laminated on a thermal release tape (Nitto NO.319Y-4LS RL) using a roller machine. The tape/graphene/copper film was put into a 2 M aqueous solution of (NH₄)₂S₂O₈ to remove the copper and subsequently washed with DI water several times before drying with N₂. The tape/graphene film was put into a 0.025 M AuCl₃/H₂O solution for 1 minute, then dried with N₂ and laminated onto PET film. Finally, the thermal release tape was released at 140 °C.

THz-TDS measurements were performed using a commercial fiber-coupled spectrometer described in detail elsewhere [5]. The samples consisting of graphene on PET were raster scanned in the focal plane of the THz beam at normal incidence to form a spatial map with a resolution of ~1 mm at 1 THz. Each measurement point in the collected maps consists of an average of 40 waveforms. The frequency-dependent sheet conductivity of graphene, \(\sigma(\omega) = \sigma_\parallel + i\sigma_\perp\), is calculated from each measurement point in a scanned map from the transmission function \(\hat{T}_{\text{film}}(\omega) = \frac{\hat{E}_{\text{film}}(\omega)}{\hat{E}_{\text{sub}}(\omega)}\) where \(\hat{E}_{\text{film}}(\omega)\) and \(\hat{E}_{\text{sub}}(\omega)\) are the Fourier transforms of the THz waveforms transmitted through graphene covered PET and non-graphene covered PET, respectively. For the PET substrate we use a complex refractive index, \(n\) [17].

The sheet resistance of 1x1 cm² graphene on PET samples was measured by 4pp measurements with probe contacts at the corners of the samples. Measurements were conducted in two configurations (A and C) [18] in order to extract the van der Pauw (vdP) sheet resistance [19]. The resistance values of the individual measurements (\(R_A\) and \(R_C\)) are used for error bars. Sheet resistance measurements on a 25x30 cm² graphene on PET sample were conducted using a commercial RTS-9 Probes Tech 4pp system.

3. Results and discussion

The sheet conductivity of graphene on ~500 μm thick high-resistivity Si is routinely determined from THz-TDS measurements by isolating the signal from one of the transients
from internal reflections in the substrate [Fig. 1(a)] by time-windowing [5,20,21]. The time between internal reflections in the PET substrate is much lower compared to standard Si substrates due to reduced thickness (∼225 μm in this case) and a lower refractive index at THz frequencies (∼1.75). Therefore, the transients from the internal reflections are all combined in a single transient as shown in Fig. 1a. \( \tilde{E}_{\text{film}}(\omega) \) and \( \tilde{E}_{\text{sub}}(\omega) \) contain terms from the directly transmitted pulse together with all the following echoes from internal reflections. The expressions for \( \tilde{E}_{\text{film}}(\omega) \) and \( \tilde{E}_{\text{sub}}(\omega) \) in such a case are

\[
\tilde{E}_{\text{film}}(\omega) = E_{1}(\omega) \frac{1}{1 - \tilde{r}_{\text{film}}(\omega) e^{-i\omega d}},
\]

\[
\tilde{E}_{\text{sub}}(\omega) = E_{1}(\omega) \frac{1}{1 - \tilde{r}_{\text{sub}}(\omega) e^{-i\omega d}},
\]

where \( \delta = \omega d n / c \) with substrate thickness \( d \). The infinite sums were simplified by the geometric series \( 1 + x + x^2 + x^3 + ... = 1/(1-x) \) for \( \text{abs}(x) < 1 \). The transmission coefficient from air to a substrate covered by a thin conducting film is

\[
\tilde{t}_{\text{film}}(\omega) = 2 / (n + Z_{t} \tilde{\sigma}(\omega)),
\]

where \( Z_{\text{t}} = 377 \, \Omega \) is the vacuum impedance. Similarly, the reflection coefficient inside a substrate covered by a thin conducting film is

\[
\tilde{r}_{\text{film}}(\omega) = (n - Z_{r} \tilde{\sigma}(\omega)) / (n + Z_{r} \tilde{\sigma}(\omega)),
\]

The transmission \( \tilde{t} \) and \( \tilde{r} \) reflection coefficients between air and non-graphene covered substrate are given by the Fresnel equations. The sheet conductivity can follow be determined from the transmission function as

\[
\tilde{\sigma}(\omega) = \frac{n_{s} e^{-n_{t} e^{-\omega d}} + (n_{s} e^{-n_{t} e^{-\omega d}} - n_{s} e^{-\omega d}) Z_{T} \tilde{T}_{\text{sub}}(\omega)}{(n_{s} + n_{t} e^{-\omega d}) Z_{T} \tilde{T}_{\text{sub}}(\omega)},
\]

where \( n_{s} = n + 1 \) and \( n_{t} = n - 1 \). The DC sheet conductance, \( \sigma_{\text{DC}} \), and the carrier scattering time, \( \tau \), can be found by fitting the real part of the frequency dependent sheet conductivity, \( \sigma_{1} \), to the Drude model [6]

\[
\tilde{\sigma}_{1}(\omega) = \frac{\sigma_{\text{DC}}}{1 - i \omega \tau}.
\]

Examples of Drude fits to \( \sigma_{1} \) for graphene on PET are shown in Fig. 1(b). The small peak in the conductivity seen at ∼0.9 THz is likely to originate from minute thickness variations in the PET substrate. The uncertainties shown for \( \sigma_{\text{DC}} \) and \( \tau \) are standard errors from the fit. It is noted that the uncertainty on the scattering time is always higher than that of the DC sheet conductance. Figure 1(b) also highlights the necessity of treating the signal as a combination of the directly transmitted and internally reflected transients. The extracted sheet conductivity is seen to have a highly oscillatory appearance if we consider the signal as only consisting of a directly transmitted transient [20,21], which would increase the uncertainty on parameters extracted by fitting to the Drude model [Eq. (4)].
THz-TDS was used to map 1x1 cm$^2$ samples of graphene on PET with 200 μm stepsize. An example of a conductivity map is shown in Fig. 2(a). The frequency dependent sheet conductivity of the central 20x20 measurement points (in order to avoid artifacts from the sample edges [22]) was fitted by the Drude model in order to extract $\sigma_{DC}$. The sheet resistance, $R_s$, was defined in each point as $1/\sigma_{DC}$. An $R_s$ histogram with the 400 measurement values is shown in Fig. 2(b). An average $R_s$ of $273 \pm 9$ Ω was determined as the mean from a generalized extreme value (GEV) distribution fitted to the histogram (black curve). The sample is electrically homogeneous with a standard deviation of less than 3.5% of the average value.

In order to validate the $R_s$ values obtained from THz-TDS measurements of graphene on PET, we conducted 4pp measurements on 7 individual 1x1 cm$^2$ samples in order to obtain the vdp-corrected sheet resistance. Figure 2(c) shows the sheet resistance obtained from the 4pp measurements compared to the $R_s$ values extracted from the central 20x20 THz-TDS measurement points on the same samples. There is a good agreement between the 4pp and THz-TDS measurements except for device 4, which also clearly has a scratch across the sample [Fig. 2(c), inset]. Such a scratch will have a big impact on the 4pp measurement as the
current flow between source and drain electrodes is affected by macroscopic disruptions [23], while the THz-TDS measurement is representing the average conductivity of the graphene covered region, and is far less sensitive to large-scale electrical discontinuity [1].

A 25x30 cm$^2$ area of graphene on PET was mapped with 1 mm stepsize by THz-TDS. Figure 3(a) shows a photograph of the sample together with the conductivity map. The sample generally seems to be homogeneously conducting except for a region in one side that appears to be scratched. The scratched region is easily identified from the THz conductivity map, while it is not easily visible by optical inspection of the sample.

We extracted $\sigma_{\text{DC}}$ and $\tau$ from each of the 34608 measurement points inside the dotted rectangle in Fig. 3(a). Histograms of $R_s$ and $\tau$ are shown in Figs. 3(b) and 3(c). From a fitted GEV distribution we find an average $R_s$ of 234 ± 4 $\Omega$. The low standard deviation ($\sim$1.7%) indicates a very high homogeneity across the whole sample. From a fitted normal distribution we find an average $\tau$ of 68.5 ± 5.6 fs.

A 25x30 cm$^2$ sample of graphene on PET grown and transferred similarly to the one shown in Fig. 3(a) was characterized by 4pp measurements. The 4pp measurements were conducted with 1 cm stepsize across the sample with a histogram of the corresponding $R_{s,4pp}$ values shown in Fig. 3(d). The average $R_{s,4pp}$ of 233 ± 10 $\Omega$ is in good agreement with the values we find from THz-TDS measurements on a similar sample highlighting both the ability of extracting $R_s$ from graphene on polymer films by THz-TDS and the homogeneity of subsequent growth, transfer, and doping processes.
The easy identification of scratches in graphene on PET samples observed in THz-TDS maps is further highlighted in Fig. 4. A sample was transported in a plastic box with indentations for holding the sample. Lines of lower conductivity with a period corresponding to the distance between indentations in the packaging are clearly visible, which emphasizes the value of THz-TDS as an efficient metrology tool for characterization of graphene on polymeric films.

![THz-TDS sheet conductivity map of σ₀ averaged from 0.3 to 1.2 THz for a 25x30 cm² graphene on PET sample that has been damaged during transport. The bottom right inset shows a photograph of the box used for packaging with indentations for holding the sample.](image)

Fig. 4. THz-TDS sheet conductivity map of $\sigma_0$ averaged from 0.3 to 1.2 THz for a 25x30 cm² graphene on PET sample that has been damaged during transport. The bottom right inset shows a photograph of the box used for packaging with indentations for holding the sample.

4. Conclusion

We have shown that THz-TDS can be used to reliably extract the conductivity of graphene on polymeric films. This greatly extends the scope and application space of non-contact electrical mapping, and provides a clear starting point for systematic optimization and consistent fabrication of large-area graphene for flexible applications, with the path towards inline process monitoring cleared. The technique has the added feature that the graphene can be measured at intermediate steps in a process, since it is non-destructive and non-contact. This is a crucial point, since testing of encapsulated films upon environmental stress factors such as changes in ambient conditions or repeated cycles of mechanical straining or bending, can be done with a high spatial resolution multiple times during such testing, to reveal the exact origin of failure and provide a deeper understanding of the interplay between the different layers in a compound material or film.

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